of anisotropic apparent $g$ values, it seems probable that the large number of lines is the result of the superposition of the hydrogen hyperfine splitting spectra of radicals oriented to give different apparent $g$ values. The presence of fewer and broader lines in the spectrum of $\mathrm{C}_{2} \mathrm{D}_{5} \mathrm{I}$ than in that of $\mathrm{C}_{2} \mathrm{H}_{5} \mathrm{I}$ (Figures 1 and 3) proves that the lines are the result of hyperfine splitting by the hydrogen atoms and is consistent with the effect of deuterium substitution on the simpler ethyl radical spectrum in the glass shown in Figure 2.
Species Responsible for Spectra. Thus far in the discussion the broad spectra have been attributed to spin-orbit coupling with iodine without considering the chemical identity of the species responsible. Iodine species having an unpaired electron which may be present in the $\gamma$-irradiated matrices giving these spectra include $\mathrm{I}, \mathrm{I}_{2}{ }^{-}, \mathrm{RCHI}, \mathrm{RI}^{+}$, and $\mathrm{RI}^{-}$. The first two of these seem to be eliminated by the isotopic substitution results which show that the observed line structure is caused by hyperfine splitting by hydrogen atoms.
One interpretation of the "momentary anneal" effects shown in Figures 4 and 5 is that a process is able to occur at the anneal temperature which converts the species responsible for the complex spectrum to that responsible for the simpler central spectrum. In the case of the polycrystalline ethyl iodide, such a process
might be $\mathrm{C}_{2} \mathrm{H}_{5} \mathrm{I}^{+}+\mathrm{I}^{-}\left(\right.$or $\left.\mathrm{C}_{2} \mathrm{H}_{5} \mathrm{I}^{-}\right) \rightarrow \mathrm{C}_{2} \mathrm{H}_{5}$ (or $2 \mathrm{C}_{2} \mathrm{H}_{5}$ ) $+\mathrm{l}_{2}$, with $\mathrm{C}_{2} \mathrm{H}_{5} \mathrm{I}^{+}$or $\mathrm{C}_{2} \mathrm{H}_{5} \mathrm{I}^{-}$being responsible for the broad spectrum. The fact that the "momentary anneal" effect occurs only with crystals frozen after supercooling and does not occur with crystals frozen slowly from a seed, or with those allowed to stand for several days prior to irradiation following freezing, suggests that it is aided by a release of strain energy or change of crystal state that occurs sharply in the narrow temperature range observed for momentary anneal.
It seems more probable that $\mathrm{C}_{2} \mathrm{H}_{5} \mathrm{I}^{+}$or $\mathrm{C}_{2} \mathrm{H}_{5} \mathrm{I}^{-}$, rather than $\mathrm{C}_{2} \mathrm{H}_{4} \mathrm{I}$, is responsible for the broad spectrum, because the latter offers less possibility for conversion to $\mathrm{C}_{2} \mathrm{H}_{5}$ during momentary anneal. Esr examination of systems in which $\mathrm{C}_{2} \mathrm{H}_{4} \mathrm{I}$ might be formed by the reactions $\mathrm{C}_{2} \mathrm{H}_{4}+\mathrm{I} \rightarrow \mathrm{C}_{2} \mathrm{H}_{4} \mathrm{I}$ or $\mathrm{C}_{2} \mathrm{H}_{3} \mathrm{I}+\mathrm{H} \rightarrow \mathrm{C}_{2} \mathrm{H}_{4} \mathrm{I}$, using H and I atoms formed by the photolysis of HI, has shown no evidence of the broad spectra. ${ }^{116}$
It is probable that the same types of trapped intermediates are formed in the alkyl iodide glasses and in the polycrystalline samples with odd carbon number as in the polycrystalline samples with even carbon number. Thus it appears that a determining factor as to whether a given trapped species produces the broad spectrum is the nature of the crystal lattice in which it resides.

# An Electron Diffraction Study of Trimethylsilyl Isothiocyanate and Trimethylsilyl Isocyanate 

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#### Abstract

The molecular structures of $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$ and $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCO}$ in the gaseous state were determined by electron diffraction using sector-microphotometer data. Neither species was found to have $\mathrm{C}_{3 v}$ symmetry. The bond distances as observed (center of gravity parameter, $r_{\mathrm{g}}$ ) and the bond angles are as follows. For $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$ $r_{g}(\mathrm{C}-\mathrm{H})=1.09 \pm 0.02, r_{\mathrm{g}}(\mathrm{C}=\mathrm{S})=1.56 \pm 0.01, r_{g}(\mathrm{~N}=\mathrm{C})=1.18 \pm 0.01, r_{g}(\mathrm{Si}-\mathrm{N})=1.78 \pm 0.02, r_{g}(\mathrm{C}-\mathrm{Si})=$ $1.87 \pm 0.01 \mathrm{~A}$, and $\angle(\mathrm{Si}-\mathrm{N}-\mathrm{C})=154 \pm 2^{\circ}$. For $\left(\mathrm{CH}_{3}\right)_{3} \operatorname{SiNCO} r_{\mathrm{s}}(\mathrm{C}-\mathrm{H})=1.10 \pm 0.02, r_{\mathrm{s}}(\mathrm{C}=\mathrm{O})=1.18 \pm$ $0.01, r_{\mathrm{s}}(\mathrm{N}=\mathrm{C})=1.20 \pm 0.01, r_{\mathrm{g}}(\mathrm{Si}-\mathrm{N})=1.76 \pm 0.02, r_{\mathrm{g}}(\mathrm{C}-\mathrm{Si})=1.89 \pm 0.01 \mathrm{~A}$, and $\angle(\mathrm{Si}-\mathrm{N}-\mathrm{C})=150 \pm 3^{\circ}$.


Microwave and electron diffraction studies of isocyanates and isothiocyanates have indicated that these molecules do not possess axial symmetry: the valence angles at the nitrogen atom are $128.1 \pm 0.5^{\circ}$ for HNCO, ${ }^{1} 130.25 \pm 0.25^{\circ}$ for HNCS, ${ }^{2} 132.25 \pm 0.25^{\circ}$ for DNCS, ${ }^{2} 125 \pm 5^{\circ}$ for $\mathrm{H}_{3} \mathrm{CNCO},{ }^{3}$ and $142^{\circ}$ for $\mathrm{H}_{3}$ CNCS. ${ }^{4}$ The more recent microwave studies of $\mathrm{H}_{3}$ CNCS, ${ }^{5} \mathrm{H}_{3} \mathrm{CNCO},{ }^{6} \mathrm{H}_{3} \mathrm{CNSO},{ }^{7}$ and $\mathrm{H}_{3} \mathrm{CNCH}_{2}{ }^{8}$

[^0]also have shown the presence of internal rotation about a single threefold axis, wherein the barriers to internal rotation of the $\mathrm{CH}_{3}$ groups are low in both $\mathrm{H}_{3} \mathrm{CNCS}$ and $\mathrm{H}_{3} \mathrm{CNSO}$, intermediate in $\mathrm{H}_{3} \mathrm{CNSO}$, and quite high in $\mathrm{H}_{3} \mathrm{CNCH}_{2}$.
The silanes appear to be different. The infrared and Raman spectra ${ }^{9}$ and the microwave spectra ${ }^{10}$ of $\mathrm{H}_{3} \mathrm{SiNCS}$ and of its deuterated compounds led to the conclusion that these species belong to the symmetry class $\mathrm{C}_{3 \mathrm{v}}$, with the heavy atoms in linear array. Also,
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Figure 1. The experimental $M_{0}(q)$ curve (top) for $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$ compared with the calculated $M_{0}(q)$ curves for $S_{154}$ (the best model shown in Table I), $S_{180}\left(\phi=130^{\circ}\right), S_{140}\left(\phi=140^{\circ}\right), S_{150}\left(\phi=150^{\circ}\right)$, $S_{160}\left(\phi=160^{\circ}\right), S_{1 \overline{7} 0}\left(\phi=170^{\circ}\right)$, and $L\left(\phi=180^{\circ}\right.$, linear).

## Experimental Section

For the diffraction experiments, the samples were purified by vacuum fractionation and checked by mass spectrometric analysis. They were maintained at 1 mm in a $12-1$. glass bulb, attached to the nozzle tube. Sectored electron diffraction photographs were taken with the apparatus previously described. ${ }^{14}$ Two sectors were used, cut for about $50-\mathrm{kv}$ electrons, to cover a range of $q$ from 10 to 45 $\mathrm{A}^{-1}[q=(40 / \lambda \sin (\theta / 2)]$, where $\lambda$ is the wavelength of the electrons, and $\theta$ is the angle of scattering. The stability of the high voltage was carefully checked during each sequence. Also, gold-foil patterns were recorded to provide scale factors. Microphotometer tracings of the diffraction patterns were taken with a Leeds and Northrup microdensitometer. The plates were oscillated about the tracing axis, which passed through the center of the pattern. Optical densities were converted into relative intensities, and the data were reduced in a manner as previously reported. ${ }^{15}$

## Analysis and Results

$\left(\mathbf{C H}_{3}\right)_{3} \mathbf{S i N C S}$. Figure 1 shows the $M_{0}(q)$ function which was obtained by combining the experimental total scattered and background intensities, corrected for nonnuclear scattering. Figure 2 is the final radial distribution curve, $f(r)$, computed with a damping factor of $\mu=0.0002303 \mathrm{~A}^{2}$ [see eq 7 in ref 15]. Figure 3 gives the numbering of the atoms and illustrates


Figure 2. The experimental radial distribution curve for $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$.
the infrared and Raman spectra of $\mathrm{Si}(\mathrm{NCO})_{4}{ }^{11}$ and $\mathrm{Si}(\mathrm{NCS})_{4}{ }^{12,13}$ have been interpreted on the basis of $\mathrm{T}_{\mathrm{d}}$ symmetry around Si , implying linear structures for the four $\mathrm{Si}-\mathrm{N}-\mathrm{C}-\mathrm{O}$ and $\mathrm{Si}-\mathrm{N}-\mathrm{C}-\mathrm{S}$ groups, respectively, However, in these and in parallel studies of $\left(\mathrm{CH}_{3}\right)_{3^{-}}$ SiNCO and $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$, Goubeau and co-workers ${ }^{12}$ concluded that the absence of splitting of the $\nu_{4}-\nu_{6}$ frequencies does not exclude an angle of $120^{\circ}$ or greater at the nitrogen atom.

We considered it important to investigate the molecular structures of these and related compounds. In this paper we report on an electron-diffraction determination of the molecular structures of $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$ and $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCO}$; a second paper will soon follow on three compounds in the class $\mathrm{Cl}_{n} \mathrm{Si}(\mathrm{NCO})_{4-n}$.
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the structure deduced for the compound. In the first step the radial distribution curve was analyzed and a molecular model was deduced using the area criterion. Theoretical intensity curves were then calculated for various models which departed somewhat from that given by the radial distribution curve and compared with the experimental intensity. The radial distribution analysis and the intensity correlations were cycled until the results became self-consistent.

In the final radial distribution curve, the first peak at 1.15 A was resolved with the aid of peak-area counts into two Gaussian peaks at 1.09 and 1.18 A corresponding to bonded $\mathrm{C}-\mathrm{H}$ and $\mathrm{C}-\mathrm{N}$ distances. The second peak was assigned to the bonded $\mathrm{C}-\mathrm{S}$ distance at 1.56 A . The third and strongest peak, at 1.85 A , was resolved into two contributions, the bonded $\mathrm{C}-\mathrm{Si}$ distance at 1.87 A and the bonded $\mathrm{Si}-\mathrm{N}$ distance at
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(15) K. Kimura and S. H. Bauer, ibid., 39, 3172 (1963).

(a)

(b)

Figure 3. The structures of (a) $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$ and (b) $\left(\mathrm{CH}_{3}\right)_{3-}$ SiNCO.
1.78 A ; the latter value was reached at the conclusion of the present analysis. A small peak at 2.44 A was assigned to the nine nonbonded $\mathrm{H} \cdots$. Si distances, and the broad curve covering the region of $r$ between 2.6 and 3.4 A may be assigned to the nonbonded $\mathrm{N} \cdots \mathrm{S}$, $\mathrm{Si} \cdots \mathrm{C}_{4}, \mathrm{~N} \cdots \mathrm{C}_{1}$, and $\mathrm{C}_{1} \cdots \mathrm{C}_{2}$ distances. Here it was reasonably assumed that the $\mathrm{N}-\mathrm{C}-\mathrm{S}$ group was linear. Significant information on the geometry of the molecular skeleton was obtained from the peak at 4.41 A which appeared fairly large, due to the nonbonded $\mathrm{Si} \cdots \mathrm{S}$ distance, without overlapping with other contributions. A value for $\phi=154^{\circ}$ was obtained for $\angle \mathrm{Si}-\mathrm{N}-\mathrm{C}_{4}$. This was the key fact which pointed to the nonlinear structure for SiNCS. It is also supported by three other radial distribution peaks around $3.70,5.03$, and 5.78 A , which may be assigned mainly to the $\mathrm{C}_{2} \cdots \mathrm{C}_{4}, \mathrm{C}_{2} \cdots \mathrm{~S}$, and $\mathrm{C}_{1} \cdots \mathrm{~S}$ distances, respectively, and give 145,150 , and $153^{\circ}$ for the $\mathrm{Si}-\mathrm{N}-\mathrm{C}_{4}$ angle, respectively. These values for the angle are, however, less reliable than that deduced from the $\mathrm{Si} \cdots \mathrm{S}$ peak at $4.41 \mathrm{~A}\left(154^{\circ}\right)$. Furthermore, the presence of a peak at about 5.03 A indicates that the molecule has a staggered rather than an eclipsed configuration of the NCS group relative to the three $\mathrm{Si}-\mathrm{Me}$ groups.

The model thus obtained was checked by correlation of intensity curves. A variety of models were tested. Various molecular intensity curves for linear structures (symmetry $\mathrm{C}_{3 \mathrm{v}}$ ) with different bond distances were computed and compared with the observed pattern. Ranges of values were selected: $1.72,1.75,1.78,1.81$, and 1.84 A for the bonded $\mathrm{Si}-\mathrm{N}$ distance, with 1.17 , 1.19 , and 1.21 A for the bonded $\mathrm{N}-\mathrm{C}$ distance. All the intensity curves computed for the linear models show unacceptable features such as: slight shoulders at $q=12$ and 48 , instead of the smooth curves which were observed; no shelves at $q=21$ and 33 instead of the recorded shelves; and a broad minimum at $q=31$ and a small bump at the minimum at $q=62$ instead of the simple minima as observed. From theoretical intensities for both staggered and eclipsed configurations with different values of $\phi$, it was found that the various configurations showed significant differences in features in the region of $q=20-30$. From the systematic comparison of the theoretical intensities with the observed intensity, it was confirmed that the nonlinear $\left(\phi=154^{\circ}\right)$, staggered configuration is the


Figure 4. The experimental $M_{0}(q)$ curve (top) for $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCO}$ compared with the calculated $M_{0}(q)$ curves for $S_{130}\left(\phi=150^{\circ}\right.$, the best model shown in Table I), $S_{1 \dot{ } 33}\left(\phi=153^{\circ}\right), S_{14 \overline{7}}\left(\phi=147^{\circ}\right)$, $E$ (eclipsed, $\phi=147^{\circ}$ ), and $L\left(\phi=180^{\circ}\right.$, linear).
best model. The calculated $M_{0}(q)$ curve for this model (Table I) is given in Figure 1 for comparison with the experimental curve. Also shown are calculated intensities for several models, $S_{130}, S_{140}, S_{150}, S_{160}, S_{170}$, and $L\left(\phi=180^{\circ}\right)$, based on the same parameters but for different values of $\phi$. From the intensity correlation, it was found that the best model, $S_{15,}$, has the values $\left(q_{\text {expt }} / q_{\text {calcd }}\right)_{\mathrm{av}}=0.9988, \sigma_{q}=0.0058,\left[M_{\text {expt }}(q) /\right.$ $\left.M_{\text {calcd }}(q)\right]_{\text {av }}=0.98$, and $\sigma_{M}=0.23$. In calculating the

Table I. Structural Parameters for $\left(\mathrm{CH}_{3}\right)_{3}$ SiNCS and $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCO}$

| Bond | $r_{\mathrm{g}}, \mathrm{A}$ | $l_{i j}, \mathrm{~A}$ |
| :---: | :---: | :---: |
| $\mathrm{C}-\mathrm{H}$ | $1.09 \pm 0.02$ | $0.080 \pm 0.010$ |
| $\mathrm{C}=\mathrm{S}$ | $1.56 \pm 0.01$ | $0.040 \pm 0.005$ |
| $\mathrm{~N}=\mathrm{C}$ | $1.18 \pm 0.01$ | $0.038 \pm 0.005$ |
| $\mathrm{Si}-\mathrm{N}$ | $1.78 \pm 0.02$ | $0.055 \pm 0.005$ |
| $\mathrm{C}-\mathrm{Si}$ | $1.87 \pm 0.01$ | $0.055 \pm 0.005$ |

$\angle \mathrm{Si}-\mathrm{N}-\mathrm{C}=\phi=154 \pm 2^{\circ}$ staggered

| $\mathrm{C}-\mathrm{H}$ | $1.10=0.02$ | $0.065 \pm 0.010$ |
| :--- | :--- | :--- |
| $\mathrm{C}=\mathrm{O}$ | $1.18 \pm 0.01$ | $0.035 \pm 0.005$ |
| $\mathrm{~N}=\mathrm{C}$ | $1.20 \pm 0.01$ | $0.035 \pm 0.005$ |
| $\mathrm{Si}-\mathrm{N}$ | $1.76=0.02$ | $0.060 \pm 0.005$ |
| $\mathrm{C}-\mathrm{Si}$ | $1.89 \pm 0.01$ | $0.060 \pm 0.005$ |

$\angle \mathrm{Si}-\mathrm{N}-\mathrm{C}=\phi=150 \pm 3^{\circ}$ staggered

Table II. Mean-Square Amplitudes for $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$ and $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCO}{ }^{6}$

|  | $\left(\mathrm{CH}_{3}\right)_{3}-$ <br> SiNCS | $\left(\mathrm{CH}_{3}\right)_{3}-$ <br> SiNCO |  | $\left(\mathrm{CH}_{3}\right)_{3}-$ <br> SiNCS | $\left(\mathrm{CH}_{3}\right)_{3}-$ <br> SiNCO |
| :--- | :---: | :---: | :--- | :---: | :---: |
| $\mathrm{C}_{1}-\mathrm{H}_{11}$ | 0.080 | 0.065 | $\mathrm{H}_{12} \cdots \mathrm{C}_{1}$ | $(0.120)$ | $(0.120)$ |
| $\mathbf{N}-\mathrm{C}_{4}$ | 0.038 | 0.035 | $\mathrm{H}_{23} \cdots \mathrm{C}_{1}$ | $(0.100)$ | $(0.120)$ |
| $\mathrm{C}_{4}-\mathrm{S}$ | 0.040 |  | $\mathrm{C}_{2} \cdots \mathrm{C}_{2}$ | $(0.080)$ | 0.080 |
| $\mathrm{C}_{4}-\mathrm{O}$ |  | 0.035 | $\mathrm{C}_{1} \cdots \mathrm{C}_{4}$ | $(0.060)$ | 0.080 |
| $\mathrm{C}_{1}-\mathrm{Si}$ | 0.055 | 0.060 | $\mathrm{C}_{2} \cdots \mathrm{~S}$ | $(0.085)$ |  |
| $\mathrm{Si}-\mathrm{N}$ | 0.055 | 0.060 | $\mathrm{C}_{2} \cdots \mathrm{O}$ |  | $(0.090)$ |
| $\mathrm{H}_{11} \cdots \mathrm{Si}$ | 0.090 | 0.080 | $\mathrm{C}_{1} \cdots \mathrm{~S}$ | $(0.065)$ |  |
| $\mathrm{C}_{1} \cdots \mathrm{~N}$ | 0.060 | 0.080 | $\mathrm{C}_{1} \cdots \mathrm{O}$ |  | $(0.090)$ |
| $\mathbf{N} \cdots \mathrm{S}$ | 0.045 |  | $\mathrm{Si}_{1} \cdots \mathrm{C}_{4}$ | 0.055 | 0.065 |
| $\mathbf{N} \cdots \mathrm{O}$ |  | 0.037 | $\mathrm{Si}_{3} \cdots \mathrm{~S}$ | 0.075 |  |
| $\mathrm{C}_{1} \cdots \mathrm{C}_{2}$ | $(0.070)$ | 0.080 | $\mathrm{Si} \cdots \mathrm{O}$ |  | 0.075 |
| $\mathrm{H}_{12} \cdots \mathrm{~N}$ | $(0.120)$ | $(0.120)$ |  |  |  |
| $\mathrm{H}_{11} \cdots \mathrm{~N}$ | $(0.100)$ | $(0.120)$ |  |  |  |

[^1]

Figure 5. The experimental radial distribution curve for $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCO}$.
theoretical intensity curves the mean-square amplitudes, $l_{i j}$, listed in Table II were used; some of these were obtained from the present radial distribution analysis and others were assumed.
$\left(\mathbf{C H}_{3}\right)_{3} \mathrm{SiNCO}$. The experimental $M_{0}(q)$ and $f(r)$ curves derived by the same procedures as described above for $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$ are shown in Figures 4 and 5, respectively. The configuration of atoms in $\left(\mathrm{CH}_{3}\right)_{3}-$ SiNCO is not so readily deduced as in $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$ because of the lower scattering power of oxygen compared to that of sulfur. However, both the analysis of the radial distribution curve and a comparison of calculated intensities with the experimental one support a similar nonlinear, staggered model for $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCO}$.
The first three peaks in $f(r)$ may be assigned to interatomic distances which are not dependent on a shape of the molecular skeleton, while the fourth peak at 3 A and the doublet near 4 A are due to distances which determine the molecular geometry. The composite features of the fourth peak cannot be accounted for on the assumption of a linear model based on the interatomic distances deduced from the first three peaks. Also the inner peak of the doublet $(\sim 3.8 \mathrm{~A})$ does not fit either the linear or the eclipsed models. Features beyond 4.2 A were not considered.

Intensity curves were calculated for a range of models; various combinations of bond distances and bond angles were tested. Some of these are shown in Figure 4. In general, it appears that linear and eclipsed models do not reproduce the intensity function in the region of $q$ from 40 to $50 \mathrm{~A}^{-1}$. The most probable value for the bond angle, $\phi=\angle \mathrm{Si}-\mathrm{N}-\mathrm{C}$, was established from consideration of values of the scale factor $\left(q_{\text {expu }} / q_{\text {calcd }}\right)_{\text {ave }}$, the index of resolution [ $\left.M_{\text {expt }}(q) / M_{\text {calcd }}(q)\right]_{\text {av }}$, and their standard deviations, $\sigma_{q}$ and $\sigma_{M}$, where $q$ in the preceding refers to $q$ values of intensity maxima and minima. The best model, $S_{150}$, has the values of $\left(q_{\text {expt }} / q_{\text {calcd }}\right)_{\text {av }}=$ $1.0012, \sigma_{q}=0.0090,\left[M_{\text {expti }}(q) / M_{\text {calcd }}(q)\right]_{\mathrm{av}}=0.98$, and $\sigma_{M}=0.21 .{ }^{16}$ The structure parameters of the best model are tabulated in Table I and shown in Figure 3. The mean amplitudes used for the intensity calculations are listed in Table II. The structure parameters for the two compounds agree with one another to within the limits of error.

[^2]
## Discussion

Among the various structures which we considered briefly and discarded without detailed analysis was one in which the $\mathrm{Si}-\mathrm{N}$ bond distance was lengthened, its axis tilted from the threefold symmetry axis, and the angle at the nitrogen atom reduced so that the longer distances remained in agreement with the well-resolved outer radial distribution peaks. Such a structure implies the presence of a three-minimum potential function for locating the NCS (or NCO) group relative to the $\mathrm{CH}_{3}$ groups. For the associated wagging motion, the barrier height should be low and the root-meansquare fluctuations in the corresponding nonbonded distances high. There was no indication of such a smearing effect in the experimental radial distribution curves.

The present electron diffraction study has led to closely similar structures for trimethylsilyl isocyanate and the thioisocyanate. The $\mathrm{Si}-\mathrm{N}-\mathrm{CX}$ atoms are not collinear, with $\angle \mathrm{Si}-\mathrm{N}-\mathrm{CS}=154^{\circ}$ and $\angle \mathrm{Si}-\mathrm{N}-\mathrm{CO}=$ $150^{\circ}$. Thus, the electron diffraction data indicate structures which were not expected by analogy based on the microwave study of $\mathrm{H}_{3} \mathrm{SiNCS},{ }^{6}$ there being substantial differences in the $\mathrm{Si}-\mathrm{N}$ bond distances and in the $\angle$ SiNC angles. Table III is a summary of reported structural parameters in various isocyanates and isothiocyanates. The bond angle at the nitrogen is less than $180^{\circ}$ in all cases except $\mathrm{Si}(\mathrm{NCO})_{4}, \mathrm{Si}(\mathrm{NCS})_{4}$, and $\mathrm{H}_{3} \mathrm{SiNCS}$. The compounds $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$ and $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCO}$, along with the others, have short $\mathrm{N}-\mathrm{C}$ bond distances ( 1.18 and 1.20 A , respectively). This may be taken as an indication of signifiant contributions by the triple-bonded Lewis resonance structures


Silicon-nitrogen bond distances show an unexpected variability. The sum of the covalent radii (Pauling ${ }^{17}$ ) is 1.87 A ; in $\mathrm{N}\left(\mathrm{SiH}_{3}\right)_{3}$ the bond distance is $1.738 \mathrm{~A},{ }^{18}$ whereas in the silyl isocyanates and thioisocyanates they range from 1.78 and 1.76 A , as reported in this
(17) L. Pauling, "The Nature of Chemical Bond" Cornell University Press, Ithaca, N. Y., 1960, p 246.
(18) K. Hedberg, J. Am. Chem. Soc., 77, 6491 (1955).

Table III. A Summary of Bond Distances and Bond Angles ( $\phi$ ) in Isocyanates and Isothiocyanates

|  | $\mathrm{Si}-\mathrm{N}$ | $\mathrm{C}-\mathrm{N}$ | Bond distance $\mathrm{N}=\mathrm{C}$ | $\text { , } \mathrm{A} \overline{\mathrm{C}=\mathrm{O}}$ | $\mathrm{C}=\mathrm{S}$ | ¢, deg. | Ref |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| HNCO |  |  | $1.207 \pm 0.01$ | $1.171 \pm 0.01$ |  | $128.1 \pm 0.5$ | 1 |
| HNCS |  |  | $1.216 \pm 0.002$ |  | $1.561 \pm 0.002$ | $130.25 \pm 0.25$ | 2 |
| DNCS |  |  | $1.216 \pm 0.002$ |  | $1.561 \pm 0.002$ | $132.25 \pm 0.25$ | 2 |
| $\mathrm{H}_{3} \mathrm{CNCO}$ |  | $\begin{gathered} 1.47 \\ \text { (assumed) } \end{gathered}$ | $1.19 \pm 0.03$ | $1.18 \pm 0.03$ |  | $125^{\circ} \pm 5$ | 3 |
| $\mathrm{H}_{3} \mathrm{CNCS}$ |  | 1.47 | 1.22 |  | 1.56 | 142 | 4 |
| $\mathrm{H}_{3} \mathrm{SiNCS}$ | $1.714 \pm 0.010$ |  | $1.211 \pm 0.010$ |  | 1.560 (assumed) | 180 | 10 |
| $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCO}$ | $1.76 \pm 0.02$ |  | $1.20 \pm 0.01$ | $1.18 \pm 0.01$ |  | $150 \pm 3^{\circ}$ | $a$ |
| $\left(\mathrm{CH}_{3}\right)_{3} \mathrm{SiNCS}$ | $1.78 \pm 0.02$ |  | $1.18 \pm 0.01$ |  | $1.56 \pm 0.01$ | $154 \pm 2$ | a |
| $\begin{aligned} & \mathrm{Si}(\mathrm{NCO})_{4} \\ & \mathrm{Si}(\mathrm{NCS})_{4} \end{aligned}$ |  |  |  |  |  | 180 | 12, 13 |

${ }^{a}$ Present work.
paper, to 1.714 A in $\mathrm{H}_{3} \mathrm{SiNCS}$, to approximately 1.65 A in the $\mathrm{Cl}_{n} \mathrm{Si}(\mathrm{NCO})_{4-n}$ sequence. ${ }^{19}$ The sensitivity of the $\mathrm{Si}-\mathrm{N}$ bond distance to the nature of the groups attached to the silicon strongly argues for the assumption that the $\mathrm{Si}-\mathrm{N}$ bond is more complex than $\sigma^{2}$ derived from an ( $\mathrm{sp}^{3}$ ) hybrid. The length of the bond appears to increase with increasing electron release by the substituent groups. In turn, the types of orbitals which determine the $\mathrm{Si}-\mathrm{N}$ bond affect the $\mathrm{Si}-$ $\mathrm{N}-\mathrm{C}$ bond angle. Ebsworth, ${ }^{20}$ in summarizing the chemical and some of the physical properties of
(19) Manuscript in preparation.
(20) E. A. V. Ebsworth, "Volatile Silicon Compounds," The MacMillan Co., New York, N. Y., 1963, Chapter 5.
silylamines, calls attention to the apparent multiplebond character of $\mathrm{Si}-\mathrm{N}$ linkages.

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# Structures of $\mathrm{C}_{7} \mathrm{H}_{10}$ Valence Tautomers as Determined by Electron Diffraction 

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#### Abstract

The structures of the valence tautomers 1,3 -cycloheptadiene and $\Delta^{6}$-bicyclo[3.2.0]heptene have been investigated in the gaseous phase by electron diffraction. Both molecules possess $\mathrm{C}_{3}$ symmetry. In 1,3 -cycloheptadiene all but one of the carbon atoms are coplanar; the C atom at the apex is tilted $73^{\circ}$ up from the plane. $\Delta^{\varepsilon \text {-Bicyclo[ } 3.2 .0] \text { heptene } \text { is in a chair conformation. For the best models the bond lengths and the bond angles }}$ are as follows. For 1,3 -cycloheptadiene the bond lengths $(\mathrm{A})$ are $\mathrm{C}_{1}=\mathrm{C}_{2}=1.35, \mathrm{C}_{2}-\mathrm{C}_{3}=1.48, \mathrm{C}_{1}-\mathrm{C}_{7}=1.54$, $\mathrm{C}_{6}-\mathrm{C}_{7}=1.55, \mathrm{C}_{1}-\mathrm{H}_{1}=1.09$, and $\mathrm{C}_{-}-\mathrm{H}_{7}=1.11 ; \angle \mathrm{C}_{3} \mathrm{C}_{2} \mathrm{C}_{1}=129^{\circ}$ and $\angle \mathrm{C}_{1} \mathrm{C}_{7} \mathrm{C}_{6}=119^{\circ}$. For $\Delta^{8}$-bicyclo[3.2.0]heptene $\mathrm{C}=\mathrm{C}=1.34, \mathrm{C}-\mathrm{C}=1.56, \mathrm{C}-\mathrm{H}=1.10 \mathrm{~A} ; \angle \mathrm{C}_{7} \mathrm{C}_{1} \mathrm{C}_{2}=105.5^{\circ}, \angle \mathrm{C}_{1} \mathrm{C}_{2} \mathrm{C}_{3}=86.7^{\circ}, \angle \mathrm{C}_{5} \mathrm{C}_{1} \mathrm{C}_{2}=109.5^{\circ}$, $\angle \mathrm{C}_{2} \mathrm{C}_{3} \mathrm{C}_{\mathrm{i}}=112.9^{\circ}, \angle \mathrm{C}_{6} \mathrm{C}_{7} \mathrm{C}_{1}=94.0^{\circ}$, and $\angle \mathrm{C}_{\mathrm{F}} \mathrm{C}_{1} \mathrm{C}_{5}=86.0^{\circ}$.


The valence tautomers 1,3 -cycloheptadiene and $\Delta^{6}$ bicyclo[3.2.0]heptene possess unexpected structural features. 1,3-Cycloheptadiene was first synthesized in 1901 by Willstatter. ${ }^{1}$ Since then there have been many investigations of this compound. Friess ${ }^{2}$ reported that it absorbs in the ultraviolet at $\lambda_{\max }=248$ $\mathrm{m} \mu$, with $\log \epsilon=3.87$, independent of the solvent used. This was confirmed by Hafner and Rellensmann, ${ }^{3}$ who

[^3]found that in $n$-hexane, $\lambda_{\text {max }}=247.4 \mathrm{~m} \mu, \log \epsilon=3.90$. The spectrum shows a single broad band which is characteristic of conjugated cyclic dienes. ${ }^{4-6}$ Raman spectra of 1,3 -cycloheptadiene ${ }^{7}$ were interpreted in terms of a cis configuration about the double bond in the ring.
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